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DESCRIPTION

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RETARDATION FILM AND OPTICAL DEVICE EMPLOYING IT

5 Technical Field

The present invention relates to a retardation film having a smaller retardation value at a shorter wavelength at a measuring wavelength of 400-700 nm, that is used in optical elements of liquid crystal display devices, anti-glare films, optical recording devices and the like, and to circular polarizing plates, elliptical polarizing plates, liquid crystal display devices and other optical devices.

Background Art

15 Retardation films are used in STN (Super Twisted Nematic) systems of liquid crystal display devices and the like, and they are employed to solve such problems as color compensation and to achieve viewing angle widening. The materials generally used in retardation films for
20 color compensation are polycarbonates, polyvinyl alcohol, polysulfone, polyethersulfone, amorphous polyolefins and the like, while the materials used in retardation films for viewing angle widening are those mentioned above, as well as polymer liquid crystals, discotic liquid
25 crystals, and the like.

A quarter-wave plate, which is one type of retardation film, can convert circularly polarized light to linearly polarized light, or linearly polarized light to circularly polarized light. This has been utilized in
30 liquid crystal display devices and, particularly, in reflective liquid crystal display devices having a single polarizing plate where the rear electrode, as viewed by an observer, is the reflecting electrode, in anti-reflection films comprising a combination of a polarizing
35 plate and a quarter-wave plate, or in combination with reflective polarizing plates composed of cholesteric liquid crystals or the like that reflect only circularly

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polarized light only in either the clockwise direction or counter-clockwise direction.

5 The retardation films used in the aforementioned single polarizing plate-type reflective liquid crystal display devices and reflective polarizing plates must have a function of converting linearly polarized light to circularly polarized light and circularly polarized light to linearly polarized light, in the visible light region with a measuring wavelength of 400-700 nm, and preferably 10 400-780 nm. When this is accomplished with a single retardation film, the retardation film ideally has a retardation of $\lambda/4$ (nm) at a measuring wavelength λ of 400-700 nm, and preferably 400-780 nm.

15 Although the aforementioned color compensating retardation film materials are commonly used as quarter-wave plates, these materials exhibit birefringent wavelength dispersion. The birefringence of most polymer films becomes larger as the measuring wavelength becomes shorter, and becomes smaller at longer wavelengths. 20 Consequently, with a single polymer film it has been difficult to achieve a smaller birefringence at shorter measuring wavelengths at a measuring wavelength λ of 400-700 nm, such as with the aforementioned ideal quarter-wave plate.

25 In order to achieve a smaller birefringence with shorter measuring wavelengths as with an ideal quarter-wave plate, Japanese Unexamined Patent Publication HEI No. 10-68816 has disclosed a technique of using a quarter-wave plate and a half-wave plate attached 30 together at an appropriate angle, and Japanese Unexamined Patent Publication HEI No. 2-285304 has disclosed a technique whereby two retardation films with different Abbe numbers are laminated.

35 Current techniques require the use of two films in order to achieve a film with a smaller retardation with shorter measuring wavelengths as with ideal quarter-wave

plates, and this has presented problems such as additional steps for film attachment and increased costs as well as greater expense for the optical design. In Japanese Unexamined Patent Publication HEI No. 3-29921 there is disclosed a retardation film obtained by uniaxially stretching a film composed of a mixture or copolymer of at least two different organic polymers, wherein the first organic polymer of the two different organic polymers has a positive photoelastic constant and the second organic polymer has a negative photoelastic constant, so that the retardation film has a larger birefringence at shorter measuring wavelengths; however, no reference is made to a method of reducing the birefringence at shorter measuring wavelengths. The present invention solves this problem by allowing realization of a retardation film with a smaller retardation, at shorter measuring wavelengths, using a single film.

Summary of the Invention

The present inventors have diligently researched polymer materials for retardation films with the aim of solving the aforementioned problem, and have succeeded in providing a retardation film comprised of a single oriented polymer film, characterized in that the retardation at wavelengths of 450 nm and 550 nm satisfies the following formulae (1) and/or (2):

$$R(450)/R(550) < 1 \quad (1)$$

$$K(450)/K(550) < 1 \quad (2)$$

where $R(450)$ and $R(550)$ represent the in-plane retardation of the oriented polymer film at wavelengths of 450 nm and 550 nm, respectively, and $K(450)$ and $K(550)$ are the values calculated by $K = [n_z - (n_x + n_y)/2] \times d$ (where n_x , n_y and n_z represent the three-dimensional refractive indexes of the oriented polymer film as the refractive indexes in the direction of the x-axis, y-axis and z-axis, respectively, and d represents the thickness

[1] A retardation film comprising a single oriented polymer film, the retardation film being characterized in that the retardation at wavelengths of 450 nm and 550 nm satisfies the following formulae (1) and/or (2), and the water absorption is no greater than 1%.

$$10 \quad K(450)/K(550) < 1 \quad (2)$$

[2] A retardation film according to [1], wherein the retardation at wavelengths of 450 nm, 550 nm and 650 nm satisfies the following formulae (3) and (4):

$$25 \quad 1.01 < R(650)/R(550) < 1.4 \quad (4)$$

[3] A retardation film according to [1] or [2], wherein the retardation is smaller with a shorter wavelength in the wavelength range of 400-700 nm.

(1) the film is composed of a polymer comprising a monomer unit of a polymer with positive refractive index anisotropy (hereunder referred to as "first monomer unit") and a monomer unit of a polymer with negative refractive index anisotropy (hereunder referred to as

"second monomer unit"),

(2) $R(450)/R(550)$ for the polymer based on the first monomer unit is smaller than $R(450)/R(550)$ for the polymer based on the second monomer unit, and

5 (3) the film has positive refractive index anisotropy.

[5] A retardation film according to [1] to [3], which comprises an oriented polymer film wherein

10 (1) the film is composed of a polymer comprising a monomer unit that forms a polymer with positive refractive index anisotropy (hereunder referred to as "first monomer unit") and a monomer unit that forms a polymer with negative refractive index anisotropy (hereunder referred to as "second monomer unit"),

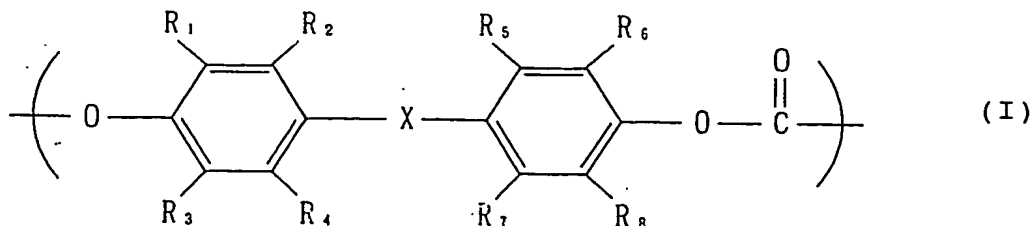
15 (2) $R(450)/R(550)$ for the polymer based on the first monomer unit is larger than $R(450)/R(550)$ for the polymer based on the second monomer unit, and

(3) the film has negative refractive index anisotropy.

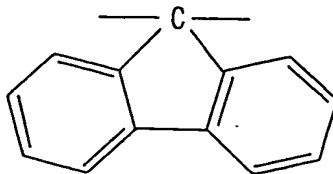
20 [6] A retardation film according to [1] to [5], wherein the oriented polymer film is a thermoplastic resin with a glass transition temperature of 120°C or higher.

25 [7] A retardation film according to [1] to [6], wherein the oriented polymer film contains a polycarbonate with a fluorene skeleton.

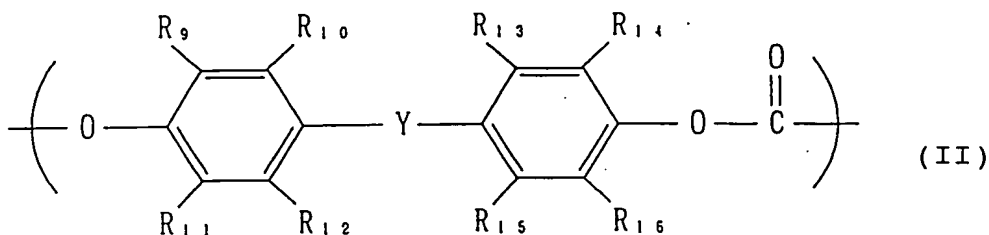
30 [8] A retardation film according to [1] to [7], which is an oriented polymer film comprising copolymer and/or blend of polycarbonates in which 30-90 mole percent of the total consists of a repeating unit represented by the following general formula (I):



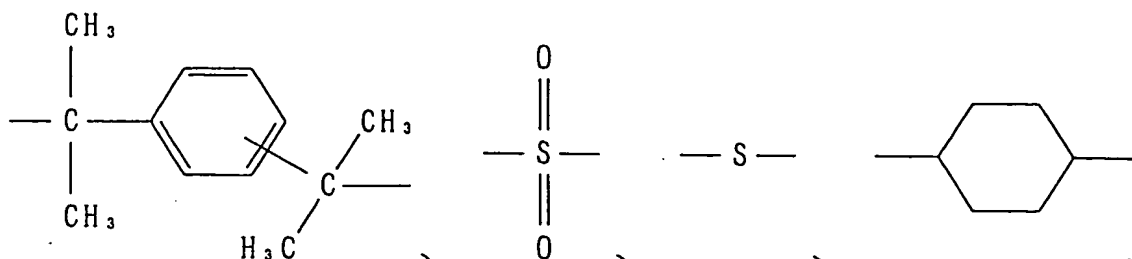
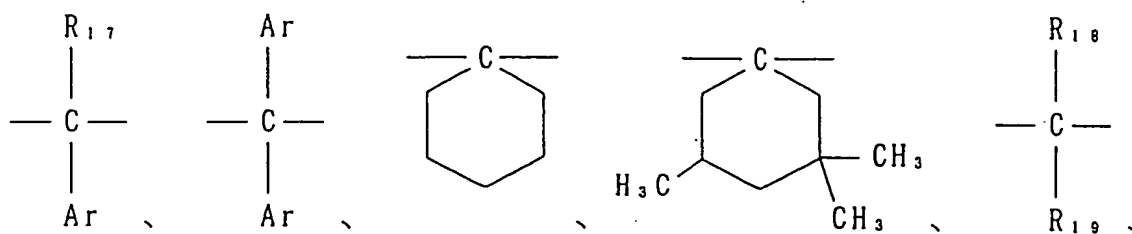
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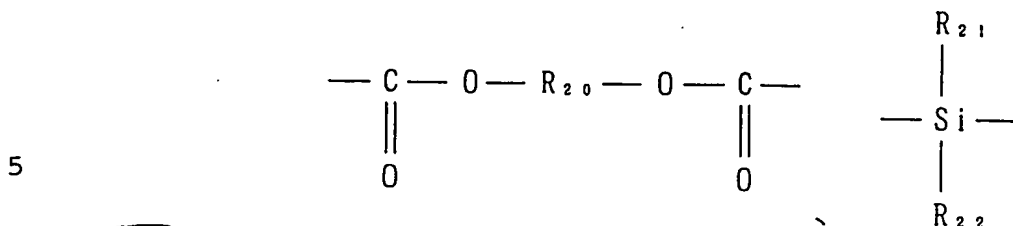
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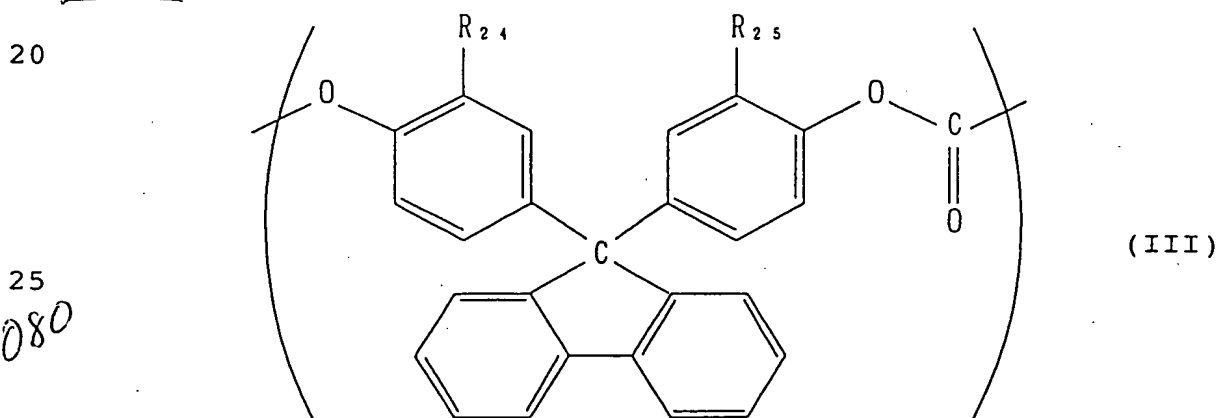
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or $\text{--- R}_{23} \text{---}$,

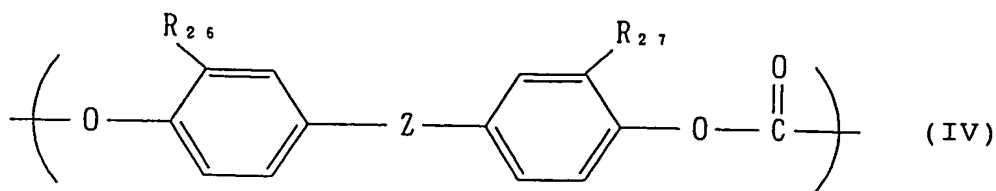
where in Y, R_{17} – R_{19} , R_{21} and R_{22} are each independently selected from among hydrogen, halogen atoms and hydrocarbon groups of 1–22 carbon atoms, R_{20} and R_{23} are selected from among hydrocarbon groups of 1–20 carbon atoms, and Ar is selected from among aryl groups of 6–10 carbon atoms.

[9] A retardation film according to [8], which is an oriented polymer film comprising copolymer and/or blend of polycarbonates in which 35–85 mole percent of the total consists of a repeating unit represented by the following general formula (III):

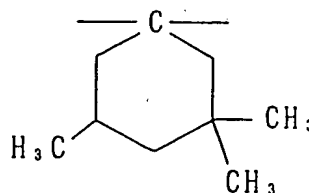
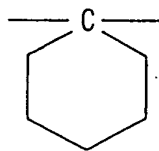
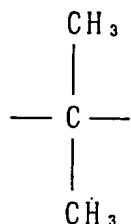


30 where R_{24} and R_{25} are each independently selected from among hydrogen and methyl, and 65–15 mole percent of the total consists of a repeating unit represented by the following general formula (IV):

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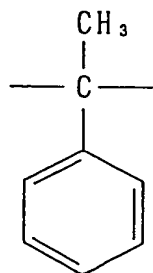


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or



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[10] A retardation film according to [5], which is a blended oriented polymer film in which the polymer with positive refractive index anisotropy is poly(2,6-dimethyl-1,4-phenyleneoxide) and the polymer with negative refractive index anisotropy is polystyrene, wherein the polystyrene content is from 67 wt% to 75 wt%.

[11] A retardation film according to [1] to [10], wherein the b* value representing the object color is 1.3 or smaller.

[12] A retardation film according to [1] to [11], which is a $\lambda/4$ plate.

[13] A retardation film according to [1] to [11], which is a $\lambda/2$ plate.

[15] A laminated retardation film prepared by laminating a $\lambda/4$ plate and a $\lambda/2$ plate, wherein both the $\lambda/4$ plate and $\lambda/2$ plate are a retardation film according to [1] to [14].

10 [17] A circular polarizing plate or elliptical
polarizing plate prepared by laminating a polarizing
plate with a retardation film according to [1] to [16].

[19] A circular polarizing plate or elliptical polarizing plate prepared by laminating a polarizing plate with a retardation film according to [1] to [16] and a reflective polarizing plate.

25 [21] A circular polarizing plate or elliptical
polarizing plate according to [20], wherein the
reflective polarizing plate is composed of a cholesteric
liquid crystal polymer.

[23] A liquid crystal display device according to [22], which is a reflective liquid crystal display device.

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$$5 \quad R(450)/R(550) < 1 \quad (1)$$

10 [26] A reflective liquid crystal display device
provided with a polarizing plate, a $\lambda/4$ plate and a
liquid crystal cell containing a liquid crystal layer
between two substrates with transparent electrodes in
that order, the reflective liquid crystal display device
15 employing as the $\lambda/4$ plate a retardation film comprising
a single oriented polycarbonate film; wherein the
retardation at wavelengths of 450 nm and 550 nm satisfies
the following formula (1):

20 where R(450) and R(550) represent the in-plane retardation of the oriented polymer film at wavelengths of 450 nm and 550 nm, respectively, and R(550) is 100-180 nm.

Figs. 1 to 4 are graphs showing the relationship between the wavelength dispersion of birefringence of a two-component blended polymer and the type and blend ratio of the polymer.

Fig. 6 is a graph showing the colorations for the laminated retardation films of Examples 16 and 17 and Comparative Examples 16 and 17.

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films, where Fig. 7 is a combination of a $\lambda/4$ plate 3 and a $\lambda/2$ plate 1, Fig. 8 is a combination of a $\lambda/4$ plate, a $\lambda/2$ plate and a polarizing plate 4, Fig. 9 is a combination of a $\lambda/4$ plate and a polarizing plate, and Fig. 10 is a combination of a polarizing plate 4, a $\lambda/4$ plate 3 and a cholesteric liquid crystal layer 5. Fig. 10 corresponds to Example 17.

Figs. 11 to 13 show examples of liquid crystal display devices, where Fig. 11 has the construction: polarizing plate 4// $\lambda/4$ plate 3//glass plate 6//liquid crystal layer 8//glass plate 6// $\lambda/4$ plate 3//polarizing plate 4//backlight system 10; Fig. 12 has the construction: polarizing plate 4// $\lambda/4$ plate 3//glass plate 6//transparent electrode 7//liquid crystal layer 8//concavoconvex reflective electrode 9//glass plate 6; and Fig. 13 has the construction: polarizing plate 4// $\lambda/4$ plate 3//light diffusing plate 11//glass substrate 6//transparent electrode 7//liquid crystal layer 8//specular reflective electrode 12//glass electrode 6. Fig. 12 corresponds to Example 13.

Fig. 14 is a graph showing the relationship between wavelength and the retardation of the retardation film of Example 1.

Figs. 15 and 16 are graphs showing the relationship between birefringent wavelength dispersion coefficient of the retardation film of Example 12 and the volume fraction of the polymer components.

Detailed Description of the Invention

The present invention, as a result of efforts to obtain an ideal $\lambda/4$ plate and $\lambda/2$ plate for a single oriented polymer film that is independent of wavelength in the visible light wavelength region, successfully provides a single oriented polymer film with a

retardation that is smaller at shorter wavelengths, thereby achieving the object described above while also providing a retardation film with properties not exhibited by the prior art.

5 The aspect of a retardation that is smaller with shorter wavelengths may be represented from a practical standpoint by the formula $R(450)/R(550) < 1$ or $K(450)/K(550) < 1$. For an extended wavelength range, $R(650)/R(550) > 1$ or $K(650)/K(550) > 1$ is preferred.
10 More preferred ranges for the retardation dispersion and K value dispersion will be explained below.

15 According to the invention, the retardation and K value of an oriented polymer film at wavelengths of 450, 550 and 650 nm will be represented as $R(450)$, $R(550)$, $R(650)$ and $K(450)$, $K(550)$, $K(650)$, respectively.

20 The retardation of an oriented polymer film is the difference in phase when light passes through a film of thickness d , based on the difference in the speeds of advance of light (refractive index) in the direction of orientation of the film and the direction normal thereto; it is known to be represented by $\Delta n \cdot d$ as the product of the difference in refractive indexes in the orientation direction and the direction normal thereto Δn and the film thickness d .

25 The orientation of an oriented polymer film according to the invention indicates a state in which the polymer molecule chains are aligned in primarily a given direction, and this state can be measured by measuring the retardation of the film ($\Delta n \cdot d$); however, the
30 orientation referred to here is one in which the retardation $R(550)$ is at least 20 nm and/or $K(550)$ is at least 20 nm, at a measuring wavelength of 550 nm. The orientation is normally produced by stretching the film.

35 Since the retardation $\Delta n \cdot d$ is proportional to the birefringence Δn for the same oriented polymer film, the

wavelength dispersion (wavelength dependency) of the retardation can be represented by the wavelength dispersion (wavelength dependency) of the birefringence Δn .

5 When the refractive index in the orientation direction in the plane of the oriented polymer film is larger than the refractive index in the direction normal thereto, the optical anisotropy is said to be positive, whereas in the opposite case the optical anisotropy is
10 said to be negative. Here, the orientation direction of the oriented polymer film is the direction of drawing, when the film has been uniaxially drawn, for example, under conditions near the glass transition temperature T_g ($T_g \pm 20^\circ\text{C}$), which are publicly known retardation film
15 manufacturing conditions. In the case of biaxial drawing, it is the direction of drawing in which the orientation is higher.

 According to the invention, reference to the retardation will mean the absolute value of the
20 retardation. In the case of negative optical anisotropy the retardation is negative, but for the purposes of the invention the positive or negative sign will be ignored unless otherwise specified.

 Also, the measuring optical wavelength used to judge
25 the positive or negative optical anisotropy will be 550 nm.

 According to the invention, it has been discovered that a retardation film comprised of a single oriented polymer film having a smaller retardation at shorter
30 wavelengths can be obtained by an oriented polymer film satisfying the following conditions (A) or (B).

 (A) An oriented polymer film wherein

 (1) the film is composed of a polymer comprising a monomer unit of a polymer with positive refractive index
35 anisotropy (hereunder referred to as "first monomer unit") and a monomer unit of a polymer with negative

(2) $R(450)/R(550)$ for the polymer based on the first monomer unit is smaller than $R(450)/R(550)$ for the polymer based on the second monomer unit, and

(B) An oriented polymer film wherein

(2) $R(450)/R(550)$ for the polymer based on the first monomer unit is larger than $R(450)/R(550)$ for the polymer based on the second monomer unit, and

Films satisfying the following conditions (C) and (D) are examples of modes that satisfy the aforementioned conditions (A) and (B).

(C) An oriented polymer film wherein

(1) the film is composed of a blend polymer comprising a polymer with positive refractive index anisotropy and a polymer with negative refractive index anisotropy and/or a copolymer comprising a monomer unit of a polymer with positive refractive index anisotropy and a monomer unit of a polymer with negative refractive index anisotropy,

(2) $R(450)/R(550)$ for the polymer with the positive refractive index anisotropy is smaller than $R(450)/R(550)$ for the polymer with the negative refractive index anisotropy, and

(3) the film has positive refractive index anisotropy.

(D) An oriented polymer film wherein

(1) the film is composed of a blend polymer comprising a polymer with positive refractive index anisotropy and a polymer with negative refractive index anisotropy and/or a copolymer comprising a monomer unit of a polymer with positive refractive index anisotropy and a monomer unit of a polymer with negative refractive index anisotropy,

(2) $R(450)/R(550)$ for the polymer with the positive refractive index anisotropy is larger than $R(450)/R(550)$ for the polymer with the negative refractive index anisotropy, and

(3) the film has negative refractive index anisotropy.

Here, a polymer with positive or negative refractive index anisotropy is a polymer that gives an oriented polymer film with positive or negative refractive index anisotropy.

The following explanation concerns the reasons for which the above retardation film is a necessary condition to have a smaller retardation with shorter measuring wavelengths.

It is generally known that the birefringence Δn of a polymer blend comprising two components, polymer A and polymer B may be represented in the following manner (H. Saito and T. Inoue, J. Pol. Sci. Part B, 25, 1629 (1987))

$$\Delta n = \Delta n_A^0 f_A \phi_A + \Delta n_B^0 f_B \phi_B + \Delta n_F \quad (a)$$

Here, Δn_A^0 is the intrinsic birefringence of polymer A, Δn_B^0 is the intrinsic birefringence of polymer B, f_A is the orientation function of polymer A, f_B is the orientation function of polymer B, ϕ_A is the volume fraction of polymer A, ϕ_B is the volume fraction of polymer B ($= 1 - \phi_A$), and Δn_F is the structural birefringence. The birefringence Δn is generally represented by $\Delta n = f\Delta n^0$. Also, Δn^0 may be determined by

combining dichromatic infrared spectroscopy with retardation measurement and the others.

Formula (a) completely ignores changes in the polarizability due to electron interaction between polymers A and B, and this assumption is also used hereunder. Also, since optical transparency is a requirement for uses of retardation films such as for the present invention, the blend is preferably a compatible blend, in which case Δn_F is sufficiently small that it may be ignored.

As concerns a retardation film with a smaller birefringence at shorter measuring wavelengths, the only measuring wavelengths considered here are 450 and 550 nm. Designating the birefringence of the retardation film at these wavelengths as $\Delta n(450)$ and $\Delta n(550)$, respectively, this is represented as $\Delta n(450)/\Delta n(550) < 1$. For a retardation film comprising a conventional polymer film, this is of course $\Delta n(450)/\Delta n(550) > 1$, and for example, $\Delta n(450)/\Delta n(550)$ is about 1.08 for a polycarbonate obtained by polymerization of bisphenol A and phosgene; it is about 1.01 even for polyvinyl alcohol which is said to have a low wavelength dispersion of birefringence.

If $\Delta n(450)/\Delta n(550)$ is defined as the birefringence wavelength dispersion coefficient, it may be represented as follows from formula (a).

$$\begin{aligned} & \Delta n(450) / \Delta n(550) \\ &= (\Delta n_A^0(450) f_A \phi_A + \Delta n_B^0(450) f_B \phi_B) / \\ & \quad (\Delta n_A^0(550) f_A \phi_A + \Delta n_B^0(550) f_B \phi_B) \quad (b) \end{aligned}$$

Here, $f_A = f_B$ is assumed because of a compatible blend, and therefore formula (b) below may be derived.

$$\begin{aligned} & \Delta n(450) / \Delta n(550) \\ &= (\Delta n_A^0(450) \phi_A + \Delta n_B^0(450) \phi_B) / \\ & \quad (\Delta n_A^0(550) \phi_A + \Delta n_B^0(550) \phi_B) \quad (c) \end{aligned}$$

The theoretical values listed in Table 1 were then

used in formula (c) to determine the birefringence wavelength dispersion values. In Table 1, the birefringence dispersion values for polymers A and B alone were listed in place of $\Delta n_A^0(450)$ and $\Delta n_B^0(450)$.

5

Table 1

Case	$\Delta n_A^0(550)$	$\Delta n_B^0(550)$	$\Delta n_A^0(450) / \Delta n_A^0(550)$	$\Delta n_B^0(450) / \Delta n_B^0(550)$
1	0.2	-0.1	1.01	1.15
2	0.2	-0.1	1.15	1.01
3	0.1	-0.2	1.01	1.15
4	0.1	-0.2	1.15	1.01

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When the values in Table 1 are inserted into formula (c), the graphs of Figs. 1-4 are obtained on the function of ϕ_A . Since polymer A is the polymer with positive refractive index anisotropy and polymer B is the negative one in Table 1, the optical anisotropy of the blend polymer is negative in the region with lower ϕ_A than the asymptotote in Figs. 1-4, while the anisotropy is positive in the region with higher ~~ϕ_A~~ than the asymptotic curve.

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As clearly seen in Figs. 1-4, in order to satisfy $\Delta n(450)/\Delta n(550) < 1$, it is necessary for the birefringence wavelength dispersion coefficient of the positive polymer to be smaller than that of the negative one and the optical anisotropy of the oriented polymer film to be positive, as in cases 1 and 3 of Table 1, or for the birefringence wavelength dispersion coefficient of the polymer alone to be larger than that of the negative one and the optical anisotropy of the oriented polymer film to be negative, as in cases 2 and 4. Here, 450 and 550 nm were used as the representative wavelengths, but the same applies even when other wavelengths are used.

In consideration of formula (c), when the positive and negative polymer birefringence wavelength dispersion

coefficients are exactly equal, a retardation film according to the invention cannot be obtained.

5 This consideration is based on formula (a), but since this idea is very well supported in actual systems such as in the examples described hereunder, the examples also substantiate the correctness of this idea.

10 While this consideration was explained for two components, the same idea applies even for three or more components. For example, in a system with two components having positive optical anisotropy and one component having negative anisotropy, the birefringence value and birefringence dispersion value of the component with positive optical anisotropy is compensated for by the volume fraction and the others between the two components with positive anisotropy, and the two components may be considered as one component when applying the idea described in the portion of the above formula (a) onward.

15 The explanation based on formula (a) concerned a blend of polymers A and B, but the same idea may be applied even to copolymers containing monomer units of different polymers, and the above-mentioned idea may be applied by considering it to comprise a homopolymer based on a first monomer unit (polymer A) and a homopolymer based on a second monomer unit different from the first monomer unit (polymer B).

25 The same idea may also be applied even to polymer blends of homopolymers and copolymers or polymer blends of two copolymers. That is, in such cases the idea may be applied by dividing the component polymers of the polymer blend into their constituent monomer units, considering the polymer blend as an aggregate of homopolymers made of respective monomer units, and considering this aggregate as a combination of polymer A comprising a group of homopolymers with positive optical anisotropy and polymer B comprising a group of homopolymers with negative anisotropy.

35 For example, for a blend of polymers X and Y having

For a homopolymer based on the first or second monomer unit, when the homopolymer is a polycarbonate, since polycarbonates are usually obtained by polycondensation of a dihydroxy compound and phosgene, from the standpoint of polymerization the monomers are the bisphenol dihydroxy compound and phosgene. In the case of such a polycarbonate, the monomer unit is said to be the portion derived from the bisphenol, and does not include the portion derived from the phosgene.

$$0.60 < R(450)/R(550) < 0.97 \quad (d)$$
$$1.01 < R(650)/R(550) < 1.35 \quad (e)$$

20

and

$$1.05 < R(650)/R(550) < 1.35 \quad (e')$$

The above is a development of the theory based on the retardation $\Delta n d$ or birefringence Δn , but the same theory can be developed based on the K value of the three-dimensional refractive indexes ($K = [n_z - (n_x + n_y)/2] \times d$). When this K value is such that $K(450)/K(550) < 1$, the wavelength dispersion is smaller with shorter wavelengths, similar to the retardation (birefringence) for light from an inclined direction that is not normal to the surface of the oriented polymer film.

The preferred ranges for the K value are those obtained by substituting K for R in the above formulae (d) and (e).

The retardation film of the invention may consist of a blend polymer or a copolymer, such as explained above.

The polymer material composing the retardation film of the invention is not particularly limited and may be any blend or copolymer that satisfies the aforementioned conditions, although preferred are materials with excellent heat resistance, satisfactory optical performance and suitability to solution film formation, with thermoplastic polymers being especially preferred. For example, one or more types may be appropriately selected from among polyacrylates, polyesters, polycarbonates, polyolefins, polyethers, polysulfine-based copolymers, polysulfone, polyethersulfone and the like. However, since problems may arise in the practicality of the retardation film if the water absorption of the oriented polymer film is greater than 1 wt%, it is important to select a film material satisfying the condition of a film water absorption of no greater than 1 wt%, and preferably no greater than 0.5 wt%.

A blend polymer must be optically transparent, and therefore it is preferably a compatible blend or one that

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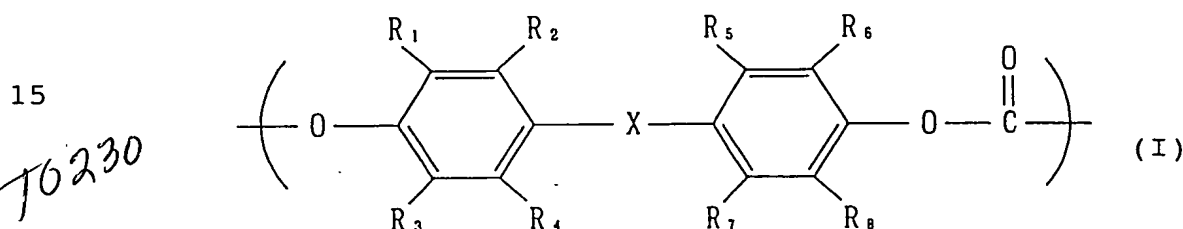
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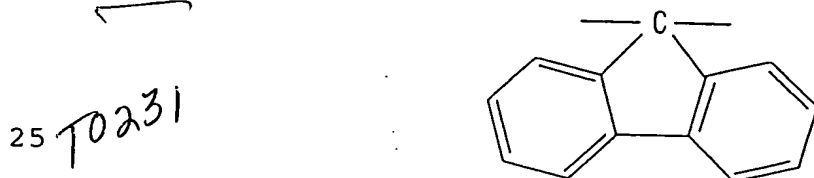
phosgene or a compound that forms a carbonic acid ester such as diphenyl carbonate, because of their excellent transparency, heat resistance and productivity.

Preferred polycarbonate copolymers are copolymers containing structures with fluorene skeletons. The component with the fluorene structure is preferably included at 1-99 mole percent.

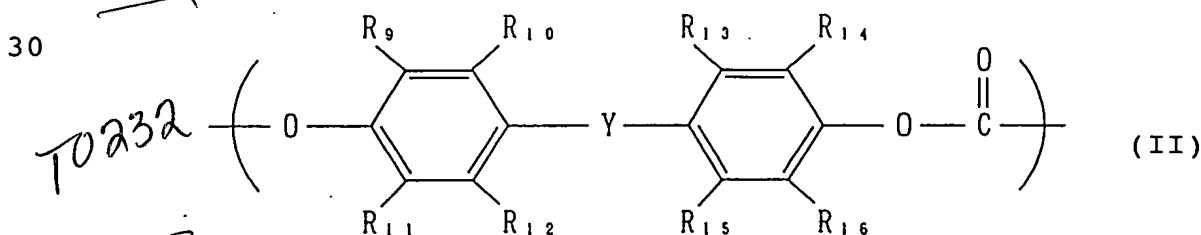
Suitable materials for the oriented polymer film of the retardation film of the invention are materials that give an oriented polymer film of a polycarbonate composed of a repeating unit represented by the following general formula (I):



20 where R_1 - R_8 are each independently selected from among hydrogen, halogen atoms and hydrocarbon groups of 1-6 carbon atoms, and X is



and a repeating unit represented by the following general formula (II):



35 where R_9 - R_{16} are each independently selected from among hydrogen, halogen atoms and hydrocarbon groups of 1-22 carbon atoms, and Y is selected from among

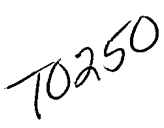
referred to as "blend polymer"). In the case of a copolymer, two or more of each of the repeating units represented by formulas (I) and (II) may be combined, and in the case of a composition as well, two or more of the
5 aforementioned repeating units may be used in combination.

In formula (I), R_1 - R_8 are each independently selected from among hydrogen, halogen atoms and hydrocarbon groups of 1-6 carbon atoms. As hydrocarbon
10 groups of 1-6 carbon atoms there may be mentioned alkyl groups such as methyl, ethyl, isopropyl and cyclohexyl, and aryl groups such as phenyl. Of these, hydrogen and methyl are preferred.

In formula (II), R_9 - R_{16} are each independently selected from among hydrogen, halogen atoms and hydrocarbon groups of 1-22 carbon atoms. As hydrocarbon
15 groups of 1-22 carbon atoms there may be mentioned alkyl groups of 1-9 carbon atoms such as methyl, ethyl, isopropyl and cyclohexyl, and aryl groups such as phenyl, biphenyl and terphenyl. Of these, hydrogen and methyl
20 are preferred.

In Y of formula (II), R_{17} - R_{19} , R_{21} and R_{22} are each independently selected from among hydrogen, halogen atoms and hydrocarbon groups of 1-22 carbon atoms, which
25 hydrocarbon groups may be the same as those mentioned above. R_{20} and R_{23} are selected from among hydrocarbon groups of 1-20 carbon atoms, which hydrocarbon groups may also be the same as those mentioned above. Ar is
30 selected from among aryl groups of 6-10 carbon atoms such as phenyl and naphthyl.

The content of formula (I), i.e. the copolymer composition in the case of a copolymer or the blend composition ratio in the case of a composition, is 30-90 mole percent of the total polycarbonate. Outside of this
35 range, the absolute value of the retardation will not be smaller with shorter wavelengths at measuring wavelengths of 400-700 nm. The content of formula (I) is preferably



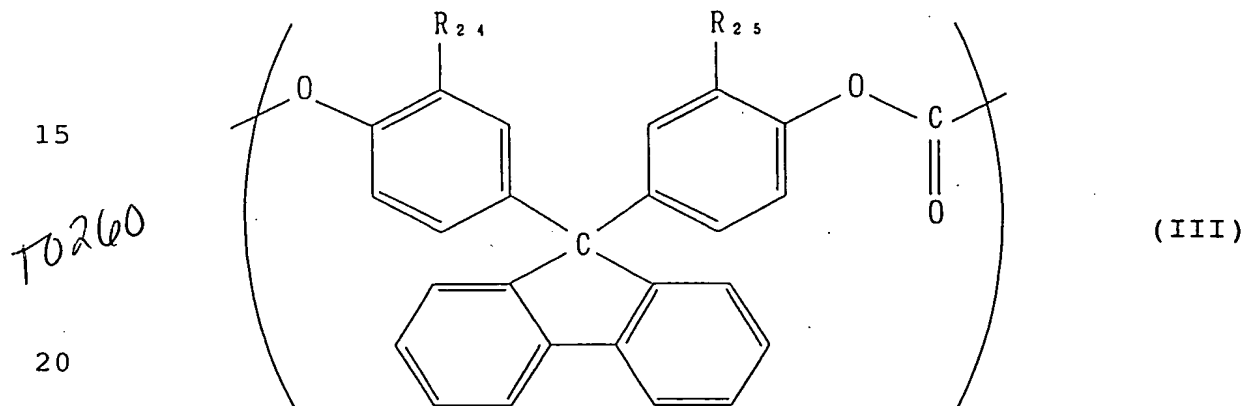
where in Y, R₁₇-R₁₉, R₂₁ and R₂₂ are each independently selected from among hydrogen, halogen atoms and hydrocarbon groups of 1-22 carbon atoms, R₂₀ and R₂₃ are each independently selected from among hydrocarbon groups of 1-20 carbon atoms, and Ar is selected from among aryl groups of 6-10 carbon atoms, such that the repeating unit represented by formula (I) constitutes 30-90 mole percent of the total polycarbonate and the repeating unit represented by formula (II) constitutes 70-10 mole percent of the whole.

30 This material is a polycarbonate copolymer comprising a repeating unit with a fluorene skeleton represented by formula (I) above and a repeating unit represented by formula (II) above, or a composition containing a polycarbonate comprising a repeating unit
35 with a fluorene skeleton represented by formula (I) above and a polycarbonate comprising a repeating unit represented by formula (II) above (hereunder also

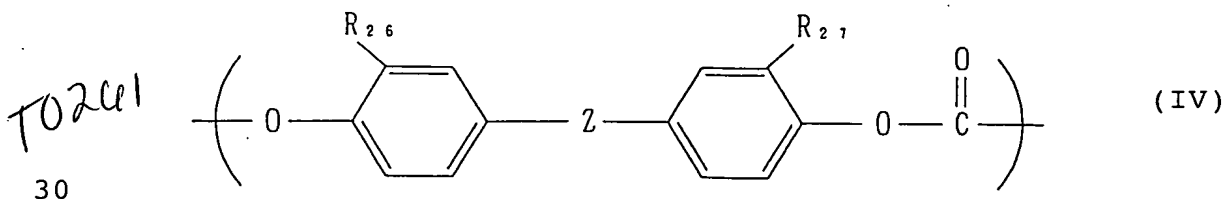
35-85 mole percent, and more preferably 40-80 mole percent, of the total polycarbonate.

Here, the above molar ratio can be determined by, for example, nuclear magnetic resonance (NMR) with the total bulk of the polycarbonate composing the oriented polymer film, regardless of the copolymer or blend polymer.

The polycarbonate in the material is preferably a polycarbonate copolymer and/or polycarbonate composition (blend polymer) comprising a repeating unit represented by the following general formula (III):



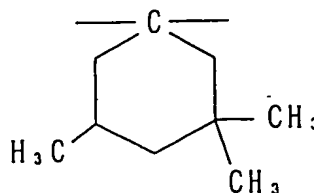
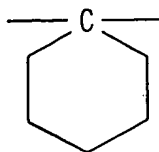
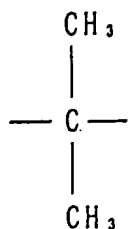
where R_{24} and R_{25} are each independently selected from among hydrogen and methyl, and a repeating unit represented by the following general formula (IV):



where R_{26} and R_{27} are each independently selected from among hydrogen and methyl, and Z is selected from among

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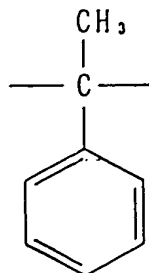
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and

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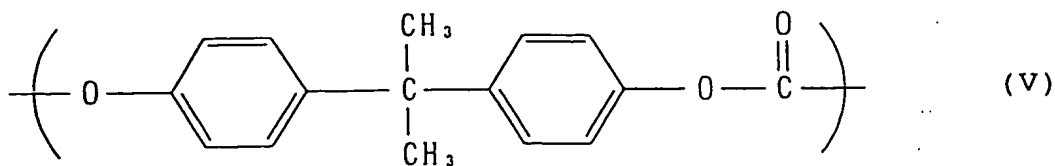
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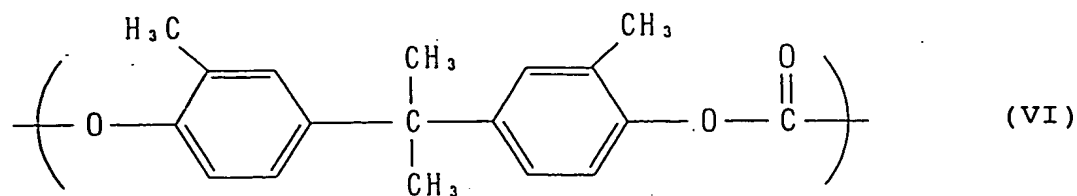
For copolymers comprising the repeating units represented by the following formulae (V) to (IX), those with repeating unit (IX) in a proportion of 40-75 mole percent are preferred; for copolymers comprising repeating units represented by the following formulae (VI) and (IX), those with (IX) in a proportion of 30-70 mole percent are preferred; for copolymers comprising repeating units represented by the following formulae (VII) and (IX), those with (IX) in a proportion of 30-70 mole percent are preferred; and for copolymers comprising repeating units represented by the following formulae (V) and (VIII), those with (VIII) in a proportion of 40-75 mole percent are preferred.

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polycarbonates, polycondensation processes and molten polycondensation processes with dihydroxy compounds and phosgene may be suitably used. For blends, compatible blends are preferred but even if they are not totally compatible the refractive indexes of the components can be matched to reduce light scattering between the components and improve the transparency.

The limiting viscosity of the material polymer of the oriented polymer film composing the retardation film of the invention is preferably 0.3-2.0 dl/g. A value below this range is a problem because the material becomes brittle and cannot maintain its mechanical strength, and a value above this range is also a problem because the solution viscosity increases excessively leading to die lines during formation of the solution film, or because purification after completion of polymerization becomes more difficult.

The retardation film of the invention is preferably transparent, preferably with a haze value of no greater than 3% and a total light transmittance of 85% or greater. The glass transition temperature of the polymer film material is preferably 100°C or higher, and more preferably 120°C or higher.

There may also be added an ultraviolet absorber such as phenylsalicylic acid, 2-hydroxybenzophenone or triphenyl phosphate, a blueing agent for adjustment of the color, an antioxidant, etc.

The retardation film of the invention employs a film of the aforementioned polycarbonate or the like that is oriented by drawing or other means. The production process for the film may be a publicly known melt extrusion process, solution casting process or the like, but solution casting is preferably used from the standpoint of film thickness irregularities and outer appearance. Suitable solvents to be used for solution casting are methylene chloride, dioxolan and the like.

Any publicly known drawing process may be used for

30 The retardation film of the invention, particularly
a single oriented polymer film, may be used to construct
a satisfactory quarter-wave plate ($\lambda/4$ plate) or half-
wave plate ($\lambda/2$ plate) with low wavelength dependency,
and for such uses it preferably satisfies $R(550) \geq 50$ nm
35 and more preferably $R(550) \geq 90$ nm; specifically, for use
as a $\lambda/4$ plate the condition $100 \text{ nm} \leq R(550) \leq 180 \text{ nm}$ is

Thus, the present invention provides, as one suitable retardation film, a retardation film consisting of a single oriented polycarbonate film, wherein the retardation at wavelengths of 450 nm and 550 nm satisfy the following formula:

where R(450) and R(550) represent the in-plane retardation of the oriented polymer film at wavelengths of 450 nm and 550 nm, respectively, and R(550) is at least 50 nm.

Cellulose acetate films that are used generally have R(550) as small as a few nm (a high acetylation degree and water absorption of about 4%) when used, for example,

for polarizing plates or support plates of optical compensation plates. For such uses, no problems arise with the retardation in practice even with orientation relaxation of R(550), and the film can therefore be used
5 as a retardation film. However, if R(550) is any larger, it is not possible to obtain a highly reliable retardation film using a cellulose acetate film. Moreover, even greater reliability is currently desired for uses inside automobiles, for example, where high
10 moisture and heat resistance are required.

The retardation film of the invention can be used as a quarter-wave plate. In this case, it is preferred to use one which has $R = \Delta n \cdot d$ equal to a quarter wavelength of 550 nm which is the highest visibility in visible
15 light.

In more general terms, to allow a single retardation film of the invention to be used as a wide-band $\lambda/4$ plate, the retardation wavelength dispersion is preferably in the ranges of

20 $0.60 < R(450)/R(550) < 0.97$ and
 $1.01 < R(650)/R(550) < 1.40$

more preferably,

$0.65 < R(450)/R(550) < 0.92$ and
 $1.03 < R(650)/R(550) < 1.30$

25 and even more preferably,

$0.70 < R(450)/R(550) < 0.87$ and
 $1.04 < R(650)/R(550) < 1.25$.

By using this type of quarter-wave plate as a single polarizing plate, for example, in a reflective liquid
30 crystal display device having a construction with a rear electrode also serving as the reflecting electrode, it is possible to obtain a reflective liquid crystal display device with excellent image quality. The retardation film can also be used on the rear side of a guest-host
35 liquid crystal layer, as viewed by an observer. The role of the retardation film in such situations is that of

converting linearly polarized light to circularly polarized light and circularly polarized light to linearly polarized light within the visible light band, and the retardation film of the invention is capable of satisfactorily achieving this purpose.

Thus, as one preferred mode of the invention there is provided a reflective liquid crystal display device having a polarizing plate, a $\lambda/4$ plate and a liquid crystal cell containing a liquid crystal layer between two substrates with transparent electrodes in that order, the reflective liquid crystal display device employing as the $\lambda/4$ plate a retardation film which is a retardation film comprising a single oriented polycarbonate film, wherein the retardation at wavelengths of 450 nm and 550 nm satisfies the following formula (1):

$$R(450)/R(550) < 1 \quad (1)$$

where $R(450)$ and $R(550)$ represent the in-plane retardation of the oriented polymer film at wavelengths of 450 nm and 550 nm, respectively, and $R(550)$ is 100-180 nm.

Such films may also be used in place of a substrate of glass or a polymer film sandwiching the liquid crystal layer of the liquid crystal display device, to serve as both the substrate and the retardation film.

An optional use is as an element that converts circularly polarized light to linearly polarized light in a reflective polarizing plate composed of cholesteric liquid crystals that reflect only circularly polarized light in either the clockwise or counterclockwise direction.

The retardation film of the invention can also be used as a circular polarizing plate or elliptical polarizing plate by attachment to a polarizing plate via a cohesive layer or adhesive layer, or the retardation film may be coated with another material for improved moisture and heat durability or enhanced solvent

The retardation film of the invention has been developed for the purpose of achieving, with a single oriented polymer film, an ideal $\lambda/4$ plate or $\lambda/2$ plate with lower birefringence at shorter wavelengths; however, since this generally provides a novel oriented polymer film with lower birefringence at shorter wavelengths and water absorption of under 1 wt%, two retardation films of the invention may be laminated together, or a retardation film of the invention may be laminated with another optical film (retardation film, polarizing plate, optical compensating plate, etc.) to fabricate a $\lambda/4$ plate or $\lambda/2$ plate that is ideal in a wider wavelength region, for example, in order to obtain a retardation film or optical film which is suitable for a wider range of uses.

20 According to another aspect of the invention there is provided a laminated retardation film prepared by laminating a $\lambda/4$ plate and a $\lambda/2$ plate, wherein both retardation films satisfy the following formulae (5) and (6).

$$1 < R(650)/R(550) < 1.4 \quad (6)$$

34

As a means of evaluating this property, it was determined whether non-colored blackness was obtained with a polarizing plate, laminated retardation film and reflective plate laminated in that order, i.e. the construction polarizing plate/laminated retardation film/reflective plate, for incident visible light rays of natural polarization from a polarizing plate. With this construction, the polarized state of the light undergoes conversion from natural polarized light → (polarizing plate) → linearly polarized light 1 → (laminated retardation film) → circularly polarized light → (reflective plate) → circularly polarized light → (laminated retardation film) → linearly polarized light 2 → (polarizing plate), and since the optical axis of the linearly polarized light 2 is shifted 90° with respect to the linearly polarized light 1, it cannot pass through the polarizing plate and the resulting appearance is black. Observation of the degree of coloration of the blackness can serve as an evaluation of the properties of the laminated retardation film.

As a result of this evaluation, non-colored blackness was obtained with a laminated retardation film comprising laminated retardation films satisfying the above formulae (5) and (6). With films not satisfying formulae (5) and (6), the laminated retardation films produced colored blackness.

Both laminated retardation films more preferably satisfy the following formulae:

$0.70 < R(450)/R(550) < 0.99$ and
 $1.01 < R(650)/R(550) < 1.30$
and even more preferably

$0.75 < R(450)/R(550) < 0.96$ and
 $1.02 < R(650)/R(550) < 1.20$.

The laminated retardation film of the invention is obtained by laminating two retardation films whose

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retardation wavelength dispersion values satisfy the above formulae (5) and (6), i.e. a half-wave plate and a quarter-wave plate, preferably in such a manner that their optical axes form an angle of from 50° to 70°.

5 When the angle of attachment is outside of this range it is not possible to obtain satisfactory properties.

The laminated retardation film can roughly achieve $R = \lambda/4$ (nm) at measuring wavelengths of 400-700 nm.

10 However, this property does not strictly have to be always ideal for a single-polarizing plate-type reflective liquid crystal display device, for example, and when incorporated into a liquid crystal display device it is important to match the liquid crystal layer or other optical members.

15 There are no particular restrictions on polymer materials to be used as retardation films so long as they satisfy the above formulae (5) and (6), and representative examples thereof were provided above; however, the aforementioned polycarbonates with fluorene
20 skeletons are most preferred. The half-wave plate and quarter-wave plate in the laminated retardation film of the invention are also preferably made of the same polymer material from the standpoint of productivity.

25 As will be demonstrated in greater detail by the examples which follow, in order to determine the results when the retardation wavelength dispersion value of the laminated retardation film does not satisfy the above formulae (5) and (6), such as when $R(450)/R(550) = 0.5$ and $R(650)/R(550) = 1.5$, a 2 x 2 optical matrix was used
30 for a simulation, the results of which are shown in Fig. 5. The calculation for the construction: polarizing plate/laminated retardation film/reflective plate is a calculation for light from the polarizing plate entering in a direction normal to the polarizing plate, and
35 exiting in the normal direction. The polarizing plate has a 100% degree of polarization and the reflective plate is an ideal specular reflection plate. The angles

resistance, or enhanced solvent resistance. In the case of a circularly polarizing plate, the order of the first retardation film in the laminated retardation film of the invention is important, and the construction: polarizing plate//half-wave plate//quarter-wave plate is necessary. The circularly polarizing plate can circularly polarize incident light in a wide wavelength range, for light incident on the polarizing plate side.

15 plate//quarter-wave plate is circularly polarized light,
in a wide wavelength range for linearly polarized light
incident to the half-wave plate side, and light exiting
it is linearly polarized light, in a wide wavelength
range for circularly polarized light incident to the
20 quarter-wave plate side.

By using this type of retardation film in a liquid crystal display device, and particularly a single polarizing plate-type reflective liquid crystal display device, it is possible to obtain a display device with excellent image quality. The reflective liquid crystal display device may be one having a construction in the order: polarizing plate, retardation film, transparent electrode-mounted substrate, liquid crystal layer, scattering reflection electrode-mounted substrate; one having a construction in the order: polarizing plate, scattering plate, retardation film, transparent electrode-mounted substrate, liquid crystal layer, specular reflection electrode-mounted substrate; one having a construction in the order: polarizing plate, retardation film, transparent electrode-mounted substrate, liquid crystal layer, transparent electrode-mounted substrate, reflective layer; or the like. As an

option, the quarter-wave plate may be used in a liquid crystal display device serving as both a transparent type and reflective type. The construction of such a liquid crystal display device may be, for example, the following: polarizing plate, retardation film, transparent electrode-mounted substrate, liquid crystal layer, reflective/transparent electrode-mounted substrate, retardation film, polarizing plate, backlight system. Also, for example, it may be used as an element to convert circularly polarized light to linearly polarized light in a reflective polarizing plate made of cholesteric crystals that reflect only circularly polarized light in either the clockwise or counter-clockwise direction, to obtain satisfactory linearly polarized light across a wide band.

The retardation film of the invention may be used as a quarter-wave plate to be used in the optical head of an optical recording device. In particular, since the retardation film can provide quarter wavelength retardation for multiple wavelengths, it can contribute to reducing the number of retardation films in an optical head employing multiple laser light sources.

Figs. 7-13 show examples of constructions of laminated retardation films and liquid crystal display devices employing retardation films according to the invention.

Examples

(Evaluation methods)

The material property values described throughout the present specification were obtained by the following evaluation methods.

(1) Measurement of R and K values

The retardation R value which is the product of the birefringence Δn and the film thickness d, and the K value which is obtained from the three-dimensional refractive indexes, were measured with the spectral

(2) Measurement of total light transmittance and haze

10 (3) Measurement of water absorption

(4) Measurement of polymer copolymerization ratio
Measurement was made by proton NMR with a "JNM-
20 alpha600" by Nippon Denshi K.K.. Particularly in the
case of a copolymer of bisphenol A and bisresolfluorene,
heavy benzene was used as the solvent and calculation was
made from the proton strength ratio for each methyl
group.

Measurement was made using a "DSC2920 Modulated DSC" by TA Instruments. The measurement was conducted not after film formation but after resin polymerization, in a flake or chip state.

An Ubbellohde viscosity tube was used to determine the limiting viscosity at 20°C in methylene chloride.

35 A "U-3500" spectrophotometer by Hitachi Laboratories
was used to determine the b* value of the L*a*b* color
specification system according to JIS Z-8729, with a 2°

ellipsometer "M150" by JASCO Corporation. The R value was measured with the incident light rays and the film surface in an orthogonal state, where $R = \Delta n \cdot d = (n_x - n_y) \cdot d$. The K value was determined by varying the angle between the incident light rays and the film surface and measuring the retardation value at each angle, calculating n_x , n_y and n_z as the three-dimensional refractive indexes by curve fitting using a publicly known refractive index ellipsoid formula, and substituting the values into the formula: $K = (n_z - (n_x + n_y)/2) \cdot d$. This requires the separate parameter of the average refractive index $n = (n_x + n_y + n_z)/3$, and this was determined by measuring the refractive index n at measuring wavelengths λ of 500, 550, 590 and 640 nm using an "Abbe refractometer 2-T" by Atago, Inc., which is an Abbe refractometer equipped with a spectrophotometric light source, and finding the refractive indexes at the other wavelengths from 4 points using the Cauchy formula ($n = a + b/\lambda^2 + c/\lambda^4$; where a , b and c are the fitting parameters). The units for both the K and R values are nm. The variables n_x , n_y and n_z are defined as follows.

n_x : Refractive index in primary stretching direction in the film plane

n_y : Refractive index in direction orthogonal to the primary stretching direction in the film plane

n_z : Refractive index in the direction normal to the film surface

(The primary stretching direction is the drawing direction in the case of uniaxial drawing, and the direction of drawing that gives the highest degree of orientation, in the case of biaxial drawing; in terms of chemical structure, it indicates the direction of orientation of the polymer main chain.)

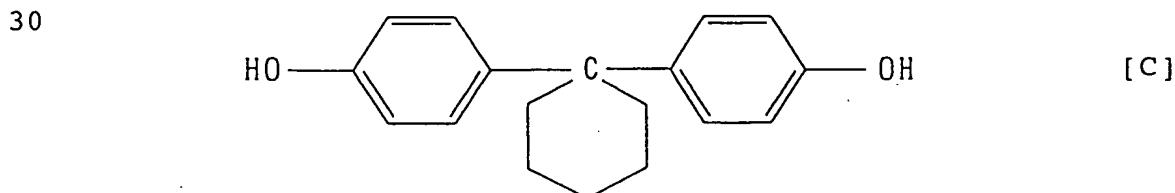
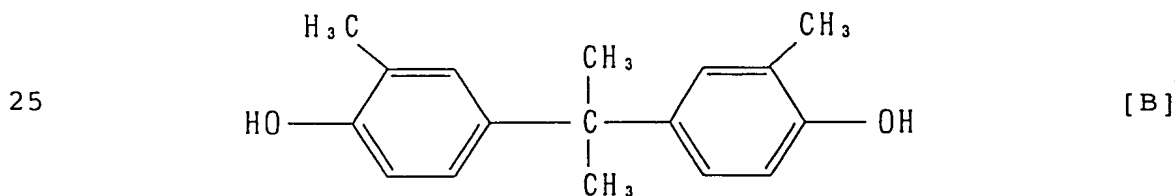
The R values at each wavelength given in the following tables are the measured values. The refractive index anisotropy is positive when $R(550) > 0$, and

(8) Measurement of film thickness

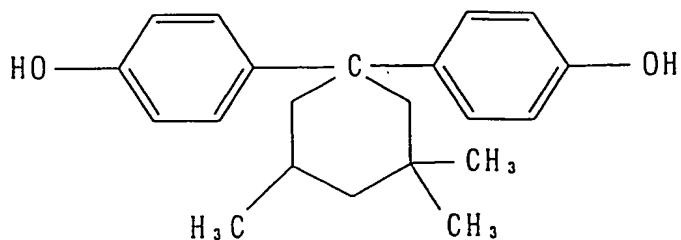
5 (9) Measurement of photoelastic constant

10 The monomer structures of the polycarbonates used in
the following examples and comparative examples are given
below.

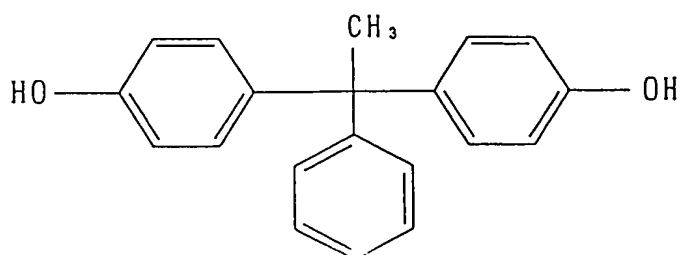
The film was placed in two constant temperature baths and allowed to stand under conditions [1] and [2] ([1] 80°C, dry, 1000 hours; [2] 60°C, 60% RH, 1000 hours), after which the film was removed and measurements (1), (2) and (7) above were repeated.



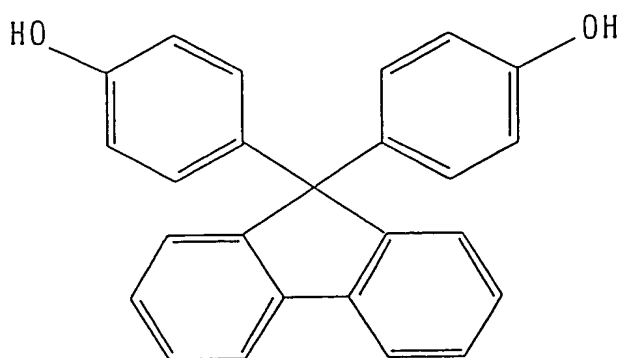
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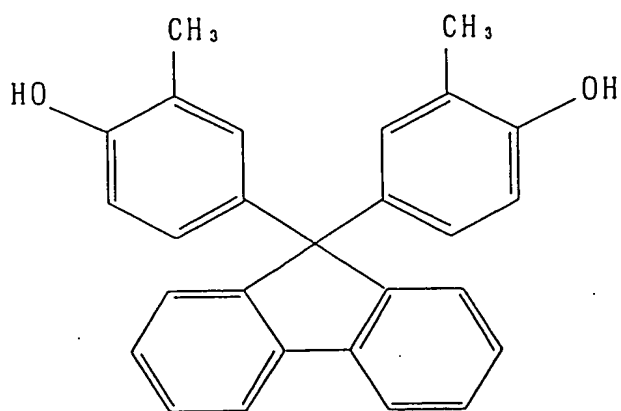
[D]



[E]



[F]



[G]

Table 3 shows the R(450)/R(550) and the refractive index anisotropy upon uniaxial stretching at near T_g, for stretched films of homopolymers of phosgene with [A] to [G] above, and polystyrene [PS] and polyphenylene oxide [PPO]. Due to the difficulty of film formation with [F], [G] or [PPO] alone, for [F] and [G] the values were

extrapolated from copolymers with small varying amounts of [A] added. For [PPO], the value was extrapolated from copolymers with small varying amounts of [PS] added.

5 Table 3

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	[A]	[B]	[C]	[D]	[E]	[F]	[G]	[PPO]	[PS]
R(450)/ R(550)	1.08	1.07	1.08	1.08	1.07	1.14	1.14	1.15	1.06
Refractive index anisotropy	+	+	+	+	+	-	-	+	-

[Example 1]

10 A sodium hydroxide aqueous solution and ion-exchange
water were charged into a reaction vessel equipped with a
stirrer, thermometer and reflux condenser, and then
monomers [A] and [G] having the structures shown above
were dissolved in the molar ratios given in Table 1 and a
15 small amount of hydrosulfite was added. After then
adding methylene chloride, phosgene was blown in at 20°C
for about 60 minutes. After adding p-tert-butylphenol
for emulsification, triethylamine was added and the
mixture was stirred at 30°C for about 3 hours to complete
20 the reaction. Upon completion of the reaction, the
organic phase was separated off and the methylene
chloride was evaporated to obtain a polycarbonate
copolymer. The compositional ratio of the obtained
copolymer was roughly the same as the monomer charging
25 ratio.

 The copolymer was dissolved in methylene chloride to
prepare a doping solution with a solid content of 15 wt%.
A cast film was formed from this doping solution, and was
subjected to free-width uniaxial stretching to a factor
30 of 1.9 at a temperature of 218°C. The solvent content of
the cast film prior to drawing was 2%, and the ratio of
the width of the film in the drawing zone to the length
in the direction of drawing was 1:1.2.

 The measurement results are summarized in Table 4.

The relationship between the retardation and the wavelength dispersion is shown in Fig. 14. The film had a smaller retardation with shorter measuring wavelength, with the largest in-plane refractive index in the direction of drawing, whereby positive refractive index anisotropy was confirmed.

The photoelastic constant of the unstretched cast film was $35 \times 10^{-13} \text{ cm}^2/\text{dyne}$.

This was followed by heat durability testing, and virtually no changes were found.

[Example 2]

The unstretched cast film prior to stretching that was fabricated in Example 1 was subjected to biaxial drawing to a factor of 1.1 at 220°C, in both the lengthwise and widthwise directions, one after the other. The measurement results are summarized in Table 4.

This was followed by heat durability testing, and virtually no changes were found.

[Example 3]

A polycarbonate copolymer was obtained by the same method as Example 1, except that the monomers listed in Table 4 were used. The compositional ratio of the resulting copolymer was roughly the same as the monomer charging ratio. A film was formed in the same manner as Example 1, and a retardation film was obtained by free-width uniaxial drawing to a factor of 1.7 at a temperature of 218°C. The solvent content of the cast film prior to drawing was 0.5%.

The measurement results are summarized in Table 4. The film had a smaller retardation with shorter measuring wavelength, and positive refractive index anisotropy was confirmed.

This was followed by heat durability testing, and virtually no changes were found.

[Example 4]

The unstretched cast film prior to drawing that was fabricated in Example 2 was subjected to biaxial drawing

This was followed by heat durability testing, and
5 virtually no changes were found.

A polycarbonate copolymer was obtained by the same method as Example 1, except that the monomers listed in Table 4 were used. The compositional ratio of the resulting copolymer was roughly the same as the monomer charging ratio. A film was formed in the same manner as Example 1, and a retardation film was obtained by free-width uniaxial drawing to a factor of 1.7 at a temperature of 218°C. The solvent content of the cast film prior to drawing was 0.2%.

This was followed by heat durability testing, and
20 virtually no changes were found.

The copolymer prepared in Example 3 was dissolved in methylene chloride so that the proportions of [A] and [G] were the same as in Example 1. The concentration of the solution was 15 wt% in terms of solid content, but it was transparent with no turbidity, and the film prepared from the solution had a haze of 0.5%, showing the two copolymers to be compatible. When the cast film was drawn under the same conditions as Example 1, the wavelength dispersion relationship for values K and R was found to be roughly equal to that in Example 1.

[Example 7]

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	Example 1	Example 2	Example 3	Example 4	Example 5	Example 7
Monomer 1 structure (charging mole%)	[A] (32)	[A] (32)	[A] (40)	[A] (40)	[A] (45)	[A] (29)
Monomer 2 structure (charging mole%)	[G] (68)	[G] (68)	[G] (60)	[G] (60)	[G] (55)	[F] (71)
T _g (°C)	227	227	218	218	213	234
Limiting viscosity [η]	0.792	0.792	0.701	0.701	0.672	0.655
R(450)	113.6	2.5	149.7	3.1	102.2	91.8
R(550)	151.0	3.3	160.3	3.1	104.3	102.1
R(650)	166.1	3.6	162.7	3.0	104.3	107.4
R(450)/R(550)	0.752	0.758	0.934	1.000	0.980	0.899
R(650)/R(550)	1.100	1.091	1.015	0.968	1.000	1.052
K(450)	-54.2	-27.2	-74.7	-60.6	-50.9	-44.1
K(550)	-74.5	-46.8	-80.2	-64.2	-50.8	-52.1
K(650)	-82.3	-67.1	-81.1	-64.4	-50.7	-53.5
K(450)/K(550)	0.728	0.581	0.931	0.944	1.002	0.846
K(650)/K(550)	1.105	1.434	1.011	1.003	0.998	1.027
Average refractive index (450)	1.635	1.635	1.627	1.627	1.628	1.633
Average refractive index (550)	1.625	1.625	1.621	1.621	1.621	1.624
Average refractive index (650)	1.621	1.621	1.615	1.615	1.615	1.622
Water absorption(%)	0.2	0.2	0.2	0.2	0.2	0.2
Haze (%)	0.7	0.9	0.9	0.8	0.9	0.9
Total light transmittance (%)	91.0	91.2	91.2	91.1	90.2	90.2
b'	0.7	0.7	0.7	0.6	1.0	1.0
Stretched film thickness (μm)	100	110	110	120	100	100

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charging ratio. A film was formed in the same manner as Example 1, and a retardation film was obtained by uniaxial drawing to a factor of 1.6 at a temperature of 210°C.

5 The measurement results are summarized in Table 5. The film had a smaller retardation with shorter measuring wavelength, and positive refractive index anisotropy was confirmed.

[Example 9]

10 A polycarbonate copolymer was obtained by the same method as Example 1, except that the monomers listed in Table 5 were used. The compositional ratio of the resulting copolymer was roughly the same as the monomer charging ratio. A film was formed in the same manner as
15 Example 1, and a retardation film was obtained by uniaxial drawing to a factor of 1.6 at a temperature of 230°C.

20 The measurement results are summarized in Table 5. The film had a smaller retardation with shorter measuring wavelength, and positive refractive index anisotropy was confirmed.

[Example 10]

25 A polycarbonate copolymer was obtained by the same method as Example 1, except that the monomers listed in Table 5 were used. The compositional ratio of the resulting copolymer was roughly the same as the monomer charging ratio. A film was formed in the same manner as Example 1, and a retardation film was obtained by stretching to a factor of 1.7 at a temperature of 230°C.

30 The optical property measurement results are summarized in Table 5. The film had a smaller retardation with shorter measuring wavelength, and positive refractive index anisotropy was confirmed.

[Example 11]

35 A polycarbonate copolymer was obtained by the same method as Example 1, except that the monomers listed in Table 5 were used. The compositional ratio of the

resulting copolymer was roughly the same as the monomer charging ratio. A film was formed in the same manner as Example 1, and a retardation film was obtained by drawing to a factor of 1.6 at a temperature of 240°C.

5 The measurement results are summarized in Table 5. The film had a smaller retardation with shorter measuring wavelength, and positive refractive index anisotropy was confirmed.

10 Table 5 Optical properties of retardation films in examples

	Example 8	Example 9	Example 10	Example 11
Monomer 1 structure (charging mole%)	[B] (60)	[C] (37)	[D] (40)	[E] (55)
Monomer 2 structure (charging mole%)	[G] (40)	[G] (63)	[G] (60)	[G] (45)
Tg (°C)	190	232	244	225
Limiting viscosity [η]	0.821	0.632	0.692	0.998
R(450)	58.5	39.1	42.0	116.6
R(550)	72.8	53.9	49.9	141.8
R(650)	80.08	59.3	52.6	150.3
R(450)/R(550)	0.803	0.725	0.841	0.822
R(650)/R(550)	1.100	1.101	1.054	1.060
K(450)	-28.0	-18.9	-20.5	-58.0
K(550)	-34.2	-26.1	-25.1	-70.3
K(650)	-40.1	-28.9	-26.7	-75.2
K(450)/K(550)	0.819	0.724	0.817	0.825
K(650)/K(550)	1.173	1.107	1.064	1.070
Average refractive index (450)	1.612	1.623	1.618	1.625
Average refractive index (550)	1.603	1.618	1.609	1.612
Average refractive index (650)	1.599	1.612	1.601	1.603
Water absorption(%)	0.2	0.2	0.2	0.2
Haze (%)	0.8	0.9	0.9	0.8
Total light transmittance (%)	90.5	90.7	90.8	90.2
b*	0.9	0.8	0.7	0.7
Stretched film thickness (μm)	80	90	90	100

[Example 12]

15 Polystyrene (obtained from Wako Junyaku Kogyo, KK.) as a polymer with negative refractive index anisotropy and polyphenylene oxide (poly(2,6-dimethyl-1,4-phenylene

oxide) obtained from Wako Junyaku Kogyo, KK.) as a polymer with positive refractive index anisotropy, were dissolved in chloroform in proportions of 70 and 30 wt%, respectively, to prepare a doping solution with a solid content of 18 wt%. A cast film was formed from the doping solution and was uniaxially drawn to a factor of 3 at a temperature of 130°C. The glass transition temperature of the film was 125°C.

The optical property measurement results are summarized in Table 6. The film had a smaller retardation with shorter measuring wavelength, and negative refractive index anisotropy was confirmed.

For reference, Fig. 15 shows the relationship between the birefringence wavelength dispersion coefficient and the polyphenylene oxide volume fraction with different blend ratios of polystyrene and polyphenylene oxide. It can be seen that the optical anisotropy is negative in the area of low polyphenylene oxide, and an area exists where the birefringence wavelength dispersion coefficient is smaller than 1. On the other hand, the value is larger than 1 in the area of high polyphenylene oxide and positive refractive index anisotropy.

Fig. 16 shows the relationship between the volume fraction and birefringence wavelength dispersion coefficient such as shown in Fig. 15, as calculated using the above-mentioned formula (c). Fig. 16 is given by using -0.10 and 0.21 as the values of the intrinsic birefringence at a wavelength of 550 nm for polystyrene and polyphenylene oxide (see D. Lefebvre, B. Jasse and L. Monnerie, Polymer 23, 706-709, 1982). Fig. 15 and Fig. 16 are well matched. The densities used of polystyrene and polyphenylene oxide were 1.047 and 1.060 g/cm³, respectively.

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backlight/cholesteric liquid crystal layer/film of Example 1/polarizing plate. The film of Example 1 functions as a $\lambda/4$ plate. The angle between the retardation axis of the film and the polarization axis was 45°. The light emitted from the polarizing plate was white with little coloration.

[Example 15]

Using an optical compensation film employing a UV-cured discotic liquid crystal layer, incorporated into the liquid crystal display device of a commercial liquid crystal monitor-equipped video camera, the discotic liquid crystal layer was peeled off from the support plate and attached to the retardation film formed in Example 2 via a cohesive layer. This was attached back to the liquid crystal display device, i.e. only the support substrate was replaced with the product of Example 2, and the device was used as a liquid crystal monitor; whereas the white display sections appeared brown-colored when viewed at an angle from the horizontal direction of the monitor in its commercial product state, the construction described here resulted in a much lower degree of coloration and hence excellent visibility. In addition, there was no loss of front contrast.

[Comparative Example 1]

A polycarbonate copolymer was obtained by the same method except that the monomers listed in Table 7 were used. The compositional ratio of the resulting copolymer was roughly the same as the monomer charging ratio. A film was formed in the same manner as Example 1, and a retardation film was obtained by drawing to a factor of 1.5 at a temperature of 240°C. The measurement results are summarized in Table 7. The film was confirmed to have a larger retardation with shorter measuring wavelength in terms of absolute value.

[Comparative Example 2]

A polycarbonate copolymer was obtained by the same method except that the monomers listed in Table 7 were

used. The compositional ratio of the resulting copolymer was roughly the same as the monomer charging ratio. A film was formed in the same manner as Example 1, and a retardation film was obtained by drawing to a factor of 1.2 at a temperature of 170°C. The measurement results are summarized in Table 7. The film was confirmed to have a larger retardation with shorter measuring wavelength in terms of absolute value.

[Comparative Example 3]

A polycarbonate copolymer was obtained by the same method except that the monomers listed in Table 7 were used. The compositional ratio of the resulting copolymer was roughly the same as the monomer charging ratio. A film was formed in the same manner as Example 1, and a retardation film was obtained by drawing to a factor of 1.5 at a temperature of 240°C. The measurement results are summarized in Table 7. The film was confirmed to have a larger retardation with shorter measuring wavelength in terms of absolute value.

[Comparative Example 4]

A polycarbonate copolymer was obtained by the same method except that the monomers listed in Table 7 were used. The compositional ratio of the resulting copolymer was roughly the same as the monomer charging ratio. A film was formed in the same manner as Example 1, and a retardation film was obtained by drawing to a factor of 1.2 at a temperature of 165°C. The measurement results are summarized in Table 7. The film was confirmed to have a larger retardation with shorter measuring wavelength in terms of absolute value.

[Comparative Example 5]

A polycarbonate copolymer was obtained by the same method except that the monomers listed in Table 7 were used. The compositional ratio of the resulting copolymer was roughly the same as the monomer charging ratio. A film was formed in the same manner as Example 1, and a retardation film was obtained by drawing to a factor of

1.5 at a temperature of 230°C. The optical property measurement results are summarized in Table 7. The film was confirmed to have a larger retardation with shorter measuring wavelength in terms of absolute value.

5 [Comparative Example 6]

10 A polycarbonate copolymer was obtained by the same method except that the monomers listed in Table 7 were used. The compositional ratio of the resulting copolymer was roughly the same as the monomer charging ratio. A film was formed in the same manner as Example 1, and a retardation film was obtained by drawing to a factor of 1.1 at a temperature of 160°C. The measurement results are summarized in Table 7. The film was confirmed to have a larger retardation with shorter measuring

15 wavelength in terms of absolute value.

[Comparative Example 7]

20 A polycarbonate copolymer was obtained by the same method except that the monomers listed in Table 7 were used. The compositional ratio of the resulting copolymer was roughly the same as the monomer charging ratio. A film was formed in the same manner as Example 1, and a retardation film was obtained by drawing to a factor of 1.3 at a temperature of 240°C. The measurement results are summarized in Table 7. The film was confirmed to have a larger retardation with shorter measuring

25 wavelength in terms of absolute value.

[Comparative Example 8]

30 A polycarbonate copolymer was obtained by the same method except that the monomers listed in Table 7 were used. The compositional ratio of the resulting copolymer was roughly the same as the monomer charging ratio. A film was formed in the same manner as Example 1, and a retardation film was obtained by drawing to a factor of 1.2 at a temperature of 175°C. The measurement results are summarized in Table 7. The film was confirmed to have a larger retardation with shorter measuring

35 wavelength in terms of absolute value.

[Comparative Example 9]

5 A polycarbonate copolymer was obtained by the same method except that the monomers listed in Table 7 were used. The compositional ratio of the resulting copolymer was roughly the same as the monomer charging ratio. A film was formed in the same manner as Example 1, and a retardation film was obtained by drawing to a factor of 1.2 at a temperature of 260°C. The measurement results are summarized in Table 7. The film was confirmed to have a larger retardation with shorter measuring wavelength in terms of absolute value.

[Comparative Example 10]

15 A polycarbonate copolymer was obtained by the same method except that the monomers listed in Table 7 were used. The compositional ratio of the resulting copolymer was roughly the same as the monomer charging ratio. A film was formed in the same manner as Example 1, and a retardation film was obtained by drawing to a factor of 1.1 at a temperature of 170°C. The optical property measurement results are summarized in Table 7. The film was confirmed to have a larger retardation with shorter measuring wavelength in terms of absolute value.

[Comparative Example 11]

25 A polycarbonate copolymer was obtained by the same method except that the monomers listed in Table 7 were used. The compositional ratio of the resulting copolymer was roughly the same as the monomer charging ratio. A film was formed in the same manner as Example 1, and a retardation film was obtained by drawing to a factor of 1.5 at a temperature of 260°C. The measurement results are summarized in Table 7. The film was confirmed to have a larger retardation with shorter measuring wavelength in terms of absolute value.

[Comparative Example 12]

35 A polycarbonate copolymer was obtained by the same method except that the monomers listed in Table 7 were used. The compositional ratio of the resulting copolymer

was roughly the same as the monomer charging ratio. A film was formed in the same manner as Example 1, and a retardation film was obtained by drawing to a factor of 1.2 at a temperature of 180°C. The measurement results are summarized in Table 7. The film was confirmed to have a larger retardation with shorter measuring wavelength in terms of absolute value.

[Comparative Example 13]

A polycarbonate copolymer was obtained by the same method except that the monomers listed in Table 7 were used. The compositional ratio of the resulting copolymer was roughly the same as the monomer charging ratio. A film was formed in the same manner as Example 1, and a retardation film was obtained by uniaxial drawing to a factor of 1.1 at a temperature of 160°C. The measurement results are summarized in Table 7. The film was confirmed to have a larger retardation with shorter measuring wavelength in terms of absolute value.

[Comparative Example 14]

A polycarbonate copolymer was obtained by the same method except that the monomers listed in Table 7 were used. The compositional ratio of the resulting copolymer was roughly the same as the monomer charging ratio. A film was formed in the same manner as Example 1, and a retardation film was obtained by uniaxial drawing to a factor of 1.1 at a temperature of 175°C. The measurement results are summarized in Table 7. The film was confirmed to have a larger retardation with shorter measuring wavelength in terms of absolute value.

[Comparative Example 15]

A polycarbonate copolymer was obtained by the same method except that the monomers listed in Table 7 were used. The compositional ratio of the resulting copolymer was roughly the same as the monomer charging ratio. A film was formed in the same manner as Example 1, and a retardation film was obtained by uniaxial drawing to a factor of 1.1 at a temperature of 170°C. The measurement

results are summarized in Table 7. The film was confirmed to have a larger retardation with shorter measuring wavelength in terms of absolute value.

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	Comp. Ex. 1	Comp. Ex. 2	Comp. Ex. 3	Comp. Ex. 4	Comp. Ex. 5	Comp. Ex. 6	Comp. Ex. 7	Comp. Ex. 8	Comp. Ex. 9	Comp. Ex. 10	Comp. Ex. 11	Comp. Ex. 12	Comp. Ex. 13	Comp. Ex. 14	Comp. Ex. 15
Monomer 1 structure (charging mole%)	[A] (19)	[A] (90)	[A] (10)	[A] (90)	[B] (19)	[B] (90)	[C] (10)	[C] (90)	[D] (10)	[D] (90)	[E] (19)	[E] (90)	[A] (100)	[C] (100)	[E] (100)
Monomer 2 structure (charging mole%)	[G] (81)	[G] (10)	[F] (90)	[F] (10)	[G] (81)	[G] (10)	[G] (90)	[G] (10)	[G] (90)	[G] (10)	[G] (81)	[G] (10)			
R(450)	-94.9	224.0	-96.4	370.1	-128.6	236.8	-57.8	285.4	-124.9	171.7	-164.7	127.3	319.5	74.6	75.3
R(550)	-68.4	210.3	-67.8	348.2	-87.9	223.4	-43.8	267.0	-95.0	160.3	-134.0	120.1	295.8	69.2	70.6
R(650)	-63.8	203.1	-62.6	335.3	-76.7	216.9	-37.4	254.5	-81.8	152.9	-116.0	116.3	282.8	66.4	68.2
R(450)/ R(550)	1.298	1.065	1.317	1.063	1.463	1.060	1.320	1.069	1.315	1.071	1.229	1.060	1.080	1.078	1.066
R(650)/ R(550)	0.873	0.966	0.855	0.963	0.872	0.971	0.854	0.953	0.861	0.954	0.866	0.968	0.956	0.959	0.965
K(450)/ K(550)	1.297	1.066	1.315	1.064	1.461	1.059	1.321	1.068	1.313	1.072	1.230	1.061	1.080	1.081	1.067
K(650)/ K(550)	0.870	0.963	0.852	0.962	0.873	0.972	0.855	0.952	0.860	0.962	0.866	0.962	0.951	0.958	0.966
Haze (%)	1.1	0.6	1.3	0.7	1.1	0.8	1.2	0.7	1.4	0.9	1.3	0.2	1.5	0.4	0.7
Total light transmittance (%)	90.0	91.8	90.2	91.5	90.3	91.7	90.1	91.1	90.8	91.3	90.2	91.3	91.0	90.7	90.7
Stretched film thickness (μm)	90	90	50	60	55	60	60	70	50	50	50	50	75	80	100

[Example 16]

A sodium hydroxide aqueous solution and ion-exchange water were charged into a reaction vessel equipped with a stirrer, thermometer and reflux condenser, and then monomers A and G having the structures shown above were dissolved in the molar ratios given in Table 8, and a small amount of hydrosulfite was added. After then adding methylene chloride, phosgene was blown in at 20°C for about 60 minutes. After adding p-tert-butylphenol for emulsification, triethylamine was added and the mixture was stirred at 30°C for about 3 hours to complete the reaction. Upon completion of the reaction, the organic phase was separated off and the methylene chloride was evaporated to obtain a polycarbonate copolymer. The compositional ratio of the obtained copolymer was roughly the same as the monomer charging ratio.

The copolymer was dissolved in methylene chloride to prepare a doping solution with a solid content of 20 wt%. A cast film was formed from this doping solution, and was subjected to uniaxial drawing to form a retardation film for $\lambda/4$ and $\lambda/2$ (nm) at a measuring wavelength of $\lambda = 550$ nm.

The film was attached at the angle shown in Table 8, to form an optical multilayer film comprising a polarizing plate//half-wave plate//quarter-wave plate//reflective plate. An adhesive was used between each optical film.

Fig. 6 shows a summary of the reflection spectrum of this optical multilayer film. There is a complete absence of coloration, and blackness with low reflectivity is achieved in the case where the reflectivity is 0 in the total wavelength range in Fig. 6; however, with the laminated retardation film of the invention, the reflectivity was lower than the comparative examples described later as seen in Fig. 6, giving a very excellent condition of blackness. The

[Example 17]

Fig. 6 shows a summary of the reflective spectrum for this optical multilayer film. As seen in Fig. 6, a very excellent condition of blackness was achieved compared to the comparative examples. The optical multilayer film was also visually examined and found to be black with no coloration.

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	Example 16	Example 17	Comp. Ex. 16	Comp. Ex. 17
Monomer 1 structure (charging mole%)	A (42)	A (34)	A (100)	-
Monomer 2 structure (charging mole%)	G (58)	G (66)	-	-
R(450) / R(550)	0.963	0.849	1.078	1.010
R(650) / R(550)	1.008	1.041	0.960	0.997
K(450) / K(550)	0.964	0.855	1.071	1.010
K(650) / K(550)	1.007	1.039	0.971	0.998
Polarizing plate polarization axis angle (°)	0	0	0	0
$\lambda/2$ plate retardation axis angle (°)	75	75	75	75
$\lambda/4$ plate retardation axis angle (°)	15	15	15	15

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Table 8 were used. A half-wave plate and quarter-wave plate were formed in the same manner as Example 16, and these were attached at the angles shown in Table 8 to prepare an optical multilayer film comprising a polarizing plate//half-wave plate//quarter-wave plate//reflective plate.

Fig. 6 shows a summary of the reflective spectrum for this optical multilayer film. The optical multilayer film was also visually examined and found to have coloration in the blackness.

[Comparative Example 17]

A half-wave plate and quarter-wave plate were formed in the same manner as Example 16, using the norbornane resin "ARTON G" by JSR, KK, and these were attached at the angles shown in Table 8 to prepare an optical multilayer film comprising a polarizing plate//half-wave plate//quarter-wave plate//reflective plate.

Fig. 6 shows a summary of the reflective spectrum for this optical multilayer film. The optical multilayer film was also visually examined and found have coloration in the blackness, compared to Examples 16 and 17.

Industrial Applicability

According to the present invention it was possible to obtain a retardation film having a smaller birefringence with shorter measuring wavelengths, even with a single film. A retardation film having such birefringence wavelength dispersion properties and a retardation to $1/4$ wavelength at a measuring wavelength of 550 nm will function as a retardation film that converts circularly polarized light to linearly polarized light and linearly polarized light to circularly polarized light in a wide wavelength range, and it can therefore be applied to a single polarizing plate- or guest/host-type reflective liquid crystal display device, or to a reflective polarizing element that reflects circularly polarized light in one direction, in order to

provide high image quality liquid crystal display devices and high performance reflective polarizing elements with satisfactory productivity.

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